

## PHYSICAL SCIENCES

# Metal nanoparticle film–based room temperature Coulomb transistor

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Single-electron transistors would represent an approach to developing less power-consuming microelectronic devices if room temperature operation and industry-compatible fabrication were possible. We present a concept based on stripes of small, self-assembled, colloidal, metal nanoparticles on a back-gate device architecture, which leads to well-defined and well-controllable transistor characteristics. This Coulomb transistor has three main advantages. By using the scalable Langmuir-Blodgett method, we combine high-quality chemically synthesized metal nanoparticles with standard lithography techniques. The resulting transistors show on/off ratios above 90%, reliable and sinusoidal Coulomb oscillations, and room temperature operation. Furthermore, this concept allows for versatile tuning of the device properties such as Coulomb energy gap and threshold voltage, as well as period, position, and strength of the oscillations.

## INTRODUCTION

In 1958, an article (1) reported on the unexpected observation of a negative differential resistance in germanium p-n junctions. What started out as an anomaly in the known current-voltage characteristics of a diode led to the development of a new technology and resulted in a Nobel Prize for the author, Leo Esaki. Even beyond the technological point of view, studying tunnel processes enables the attainment of fundamental physical knowledge as Esaki mentioned in his Nobel Prize lecture (2). Today, especially single-electron applications (3, 4) are a proof of how quantum mechanical tunneling can enable new functionalities in conventional devices (5). Despite being similar to a common field-effect transistor, the single-electron transistor (SET) does not rely on the semiconductor band gap but instead on the Coulomb energy gap. The resulting transfer characteristics exhibit periodic on and off states known as Coulomb oscillations, which might render new applications possible in the future. The working principle of the SET is well known both experimentally (6–11) and theoretically (12) and has been implemented in silicon devices (13–16). Nevertheless, using this effect in an actual device that not only yields reproducible and predictable results but can also be fabricated in an inexpensive, easy, and industrially applicable way has remained a challenge. Wet-chemical synthesis is the best-suited approach to preparing monodisperse nanoparticles with tunable properties such as size and interparticle distance. They are solution-processable, stable, and unstrained (17, 18). However, reliably contacting single or multiple particles is not easily achieved. For example, bifunctional linker molecules (19, 20), electrostatic trapping (21), and induced dipoles (22) are known approaches to catching a single nanoparticle or even assemblies thereof between readily fabricated electrodes and to demonstrating SET function (23, 24). In semiconductor-nanoparticle arrays, the coherence of the superlattice determines whether Coulomb charging or band-like transport governs the electrical transport (25). In extended metallic nanoparticle films, the presence of the effect of Coulomb blockade can be extracted by analysis of the transport characteristics (26, 27). To include high-quality nanoparticles as a two-dimensional (2D) array into an electrical device, we use a unique combination of both top-down and bottom-up techniques. By means of the Langmuir-Blodgett method (28–33), the particles' tendency

to self-assemble is exploited to create an ordered, homogeneous monolayer of evenly spaced nanoparticles. The position, shape, and size of the monolayer can be easily and precisely controlled by a resist mask. With the Langmuir-Blodgett method as a link between bottom-up and top-down, we can take advantage of the reliability and accessibility of physical techniques, such as lithography and sputter deposition, for the general device fabrication while still profiting from the chemically synthesized and uniformly assembled high-quality nanoparticles. Using confined metal nanoparticle films as a conductive channel in field-effect transistor geometry allows for inducing Coulomb oscillations via a gate electrode. This work demonstrates an alternative device concept that renders thorough and systematic investigations possible. Understanding parameter dependencies and influences allows for tailoring the devices to the needs of future applications and improvement in performance. We present Coulomb transistors with on/off ratios of more than 90%, highly tunable properties, and an unprecedented clarity of Coulomb oscillations up to room temperature.

## RESULTS AND DISCUSSION

### Device preparation

Colloidal cobalt-platinum (CoPt) nanoparticles (fig. S1) with diameters of  $(2.3 \pm 0.2)$  nm,  $(3.5 \pm 0.3)$  nm, and  $(4.0 \pm 0.3)$  nm were synthesized via the hot injection method. CoPt was chosen because of the highly tunable and well-established synthesis (34) and because the high amount of platinum in the particles avoids oxidation and electromigration. The particle size can easily be adapted by varying the ratio of the platinum source and the amine that is used as one of two ligands. These ligands keep the particles solution-processable, and their chain length defines the particle spacing and thus the tunnel barrier width. In our case, oleylamine and oleic acid were mainly used as ligands resulting in a particle spacing of about 1.9 nm (medium and small particles). Good results have also been obtained using the shorter-chained versions decylamine and decanoic acid that lead to a separation of 1.4 nm (large particles). The electrodes were prepared on Si/SiO<sub>2</sub> substrates by a combination of electron-beam lithography including a marker-alignment procedure, metal evaporation of titanium and gold, and sputter deposition of additional SiO<sub>2</sub> as gate dielectric (refer also to Materials and Methods). The nanoparticle array was prepared and included onto the sample geometry by the Langmuir-Blodgett method (setup in fig. S2) (28–33). For this, the particles are casted onto a liquid subphase and

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carefully forced together by barriers until they constitute a densely packed, hexagonally ordered monolayer (30, 35), which is monitored via in situ surface-pressure measurements. Lifting the former submerged substrate and thereby collecting the particles on the surface transfers the monolayer onto the devices. The electrical characterization was performed under vacuum atmosphere in a closed cycle cryogenic probe station at temperatures between 4.5 K and room temperature. An exemplary nanoparticle monolayer, micrographs of the final devices, and a scheme can be found in Fig. 1.

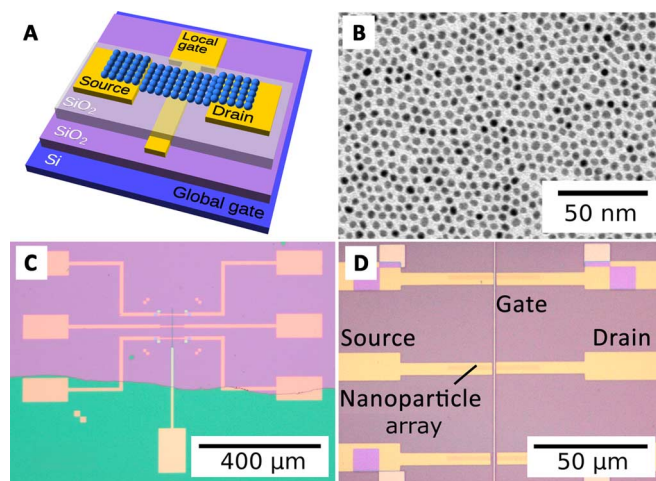
This fabrication route is not only simple and scalable but also especially highly tunable. On the one hand, the particles can be adjusted in size and shape as well as their interparticle spacing (17, 18, 30, 32). On the other hand, the monolayer can be patterned by lithography to define stripe-shaped nanoparticle arrays. Adjusting the length and width controls the total current as expected according to the varying number and length of possible conduction paths. Using this stripe-shaped monolayer channel allows full electrostatic control over all particles at once from the perpendicular direction without screening or stray currents (33). However, beyond that, the new device concept exposes further dependencies that can be used to adjust also working bias or periodicity: A local back-gate geometry is used, which allows fabrication of all electrodes before particle deposition. Consequently, the pristine ordering in the monolayer is preserved, and interferences due to further preparation steps or trap states in a capping dielectric can be avoided (figs. S3 and S4). Because of the increased quality of transfer characteristics, it was possible to observe clear Coulomb oscillations at about 210 K with the very same particles as in the study of Lehmann *et al.* (33), where 150 K was the highest temperature. Furthermore, this enables a systematic analysis of the parameters' influence on the device performance. Among others, the effect of temperature, lateral bias voltage, dielectric thickness, and gate position on the transport and on the device performance has been investigated.

### Coulomb blockade and electrical transport

Applying a bias voltage to the nanoparticle channel shows that conduction in this system is not purely metallic but is governed by tunnel barriers between individual particles. The transport is hindered due to the potential barrier given by the charging energy (12)

$$E_C = \frac{e^2}{2C} \quad (1)$$

required to overcome the electrostatic repulsion when an additional electron transfers onto a particle. The high charging energy results from the particles' inherent small self-capacitance  $C$ , which is proportional to the particle radius  $r$ . A detailed discussion of the capacitance and an estimation of its different contributions can be found in text S1. From this, a charging energy of about  $E_C = 100$  meV is calculated for the medium-sized particles. During transport, this Coulomb blockade leads to the emergence of discrete energy levels separated by a Coulomb energy gap of  $2 \cdot E_C$ . Charge carriers can tunnel through the barriers to percolate through the monolayered channel. However, elastic tunneling is highly sensitive to the energy difference between the initial and final states (36) and thus to the electrons' energy given by the bias voltage and temperature. Synthesizing the nanoparticles by colloidal chemistry allows for a precise tailoring of the system's energy levels. Very small and uniform capacitances can be achieved with monodisperse nanoparticles. The tunnel barrier width can be adjusted through the interparticle spacing defined by the organic ligands.



**Fig. 1. Device design.** (A) Schematic drawing. (B) Transmission electron micrograph of the CoPt nanoparticle monolayer as assembled by the Langmuir-Blodgett method. (C and D) Light micrographs of the final devices with three pairs of source and drain contacts as well as a local back-gate electrode. The gap between the leads and thus the length of the nanoparticle channel is 1, 2, or 3  $\mu\text{m}$  (top to bottom).

Thorough investigations have been conducted with the medium-sized particles of 3.5 nm diameter and will be presented in the following if not stated otherwise. Figure 2 (A and B) displays the output characteristics of an exemplary device. For low temperatures (compare the lower inset in Fig. 2A), no transport takes place up to a certain voltage that delivers the energy needed to overcome the Coulomb blockade. This so-called threshold voltage decreases with increasing temperature, which is very well visible in the logarithmic plot of the respective differential conductances in Fig. 2B. At around 90 K (bold green line in Fig. 2, A and B), there is no observable Coulomb blockade anymore as can be seen from the appearance of a finite zero-bias conductance. Even at room temperature, the characteristic is still slightly nonlinear. The monolayer system is well suited to investigate the temperature transition of the transport mechanism (31–33). In accordance with previous observations, electrical transport at low temperatures takes place mainly through tunneling (direct or Nordheim-Fowler tunneling), where the current is strongly dependent on the source-drain voltage (37). At elevated temperatures, transport is increasingly due to thermally activated processes like electron hopping. The current-voltage dependencies of both mechanisms are exemplarily fitted to the data in the two insets in Fig. 2A (more detailed in fig. S5).

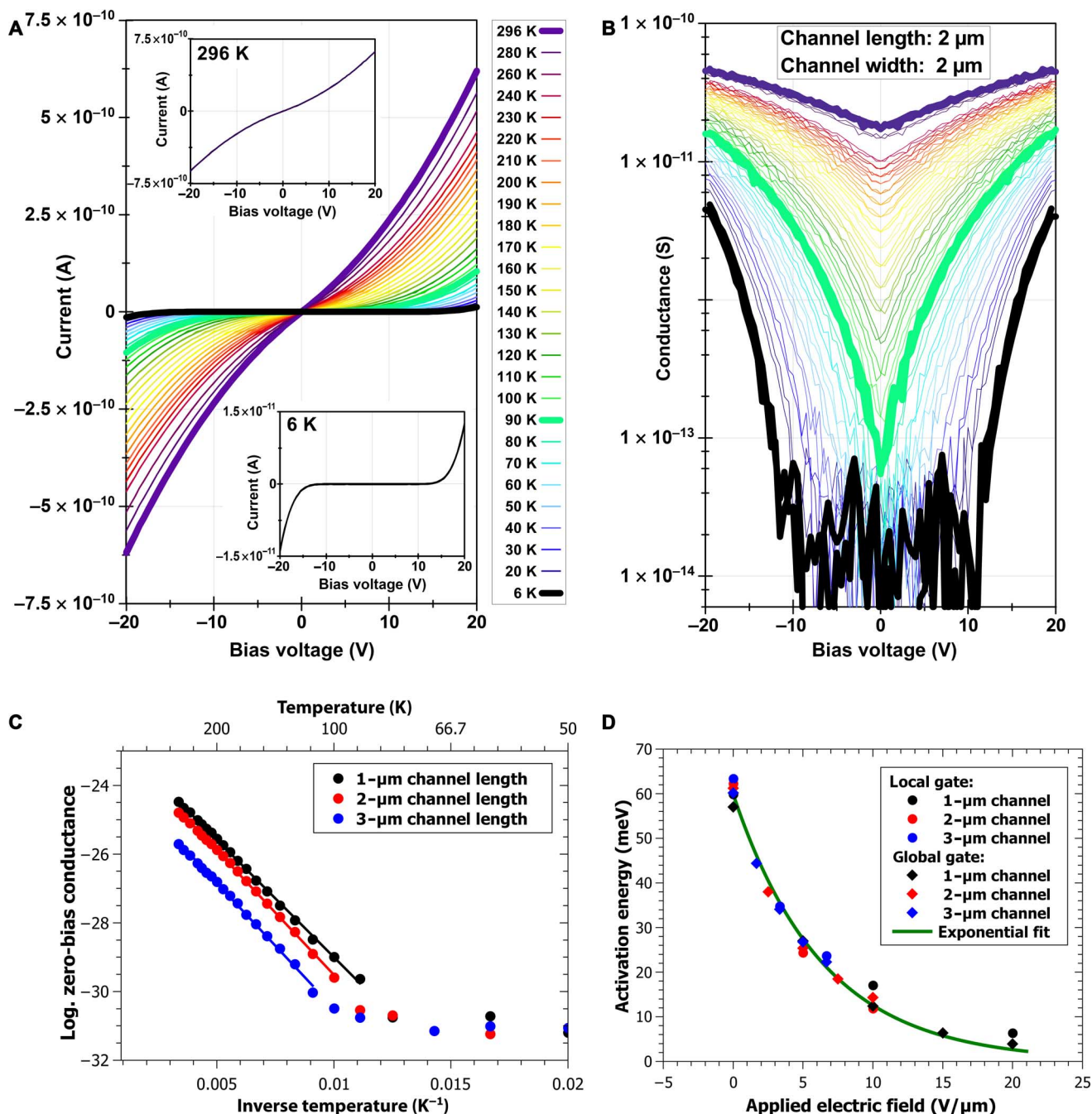
In addition, the analysis of the temperature dependency of the differential conductance  $G$  can give further information. The zero-bias conductance at temperatures  $T$  above the Coulomb blockade regime follows the equation

$$G \propto \exp\left(\frac{1}{Tv}\right) \quad (2)$$

The parameter  $v$  can take different values depending on the underlying transport type. Although  $v = 1/3$  and  $v = 1/2$  would suggest variable-range hopping (38, 39), our data are best fitted with  $v = 1$ . This Arrhenius-type behavior can be seen in Fig. 2C and indicates nearest-neighbor hopping. In the case of Arrhenius behavior, the differential zero-bias conductance

$$G \propto \exp\left(-\frac{E_A}{k_B T}\right) \quad (3)$$

can be used to determine the system's activation energy  $E_A$ , where  $k_B$  is the Boltzmann constant. For the medium-sized particles ( $r = 1.75$  nm),



**Fig. 2. Output characteristics.** (A) Current-voltage curves for different temperatures show the transition from a system governed by Coulomb blockade to almost ohmic conduction. Lowest and highest temperatures are included as insets to provide a good comparability and are fitted with the respective transport mechanism. (B) Same data plotted logarithmically as differential conductance over bias voltage. The decrease of the blockade-regime width, changes in threshold voltage, and increase in zero-bias conductance (at 90 K and above) can be seen. (C) The Arrhenius plot allows for a linear fit of the logarithmic zero-bias conductance outside of the Coulomb blockade regime. The slope indicating the activation energy is similar for all channel lengths. (D) The activation energy decreases exponentially with the applied electric field and is not dependent on channel length or gating geometry.

this yields  $E_A = 61$  meV. The dependency on the particle size will be discussed later. Applying a lateral bias voltage will lower the effective barrier for the charge carriers and thus the activation energy. Using the differential conductance at nonzero source-drain voltage values and fitting the temperature dependency as before allows for obtaining these activation energies. As shown in Fig. 2D, the activation energy decays roughly exponentially with the applied electric field independent of channel length

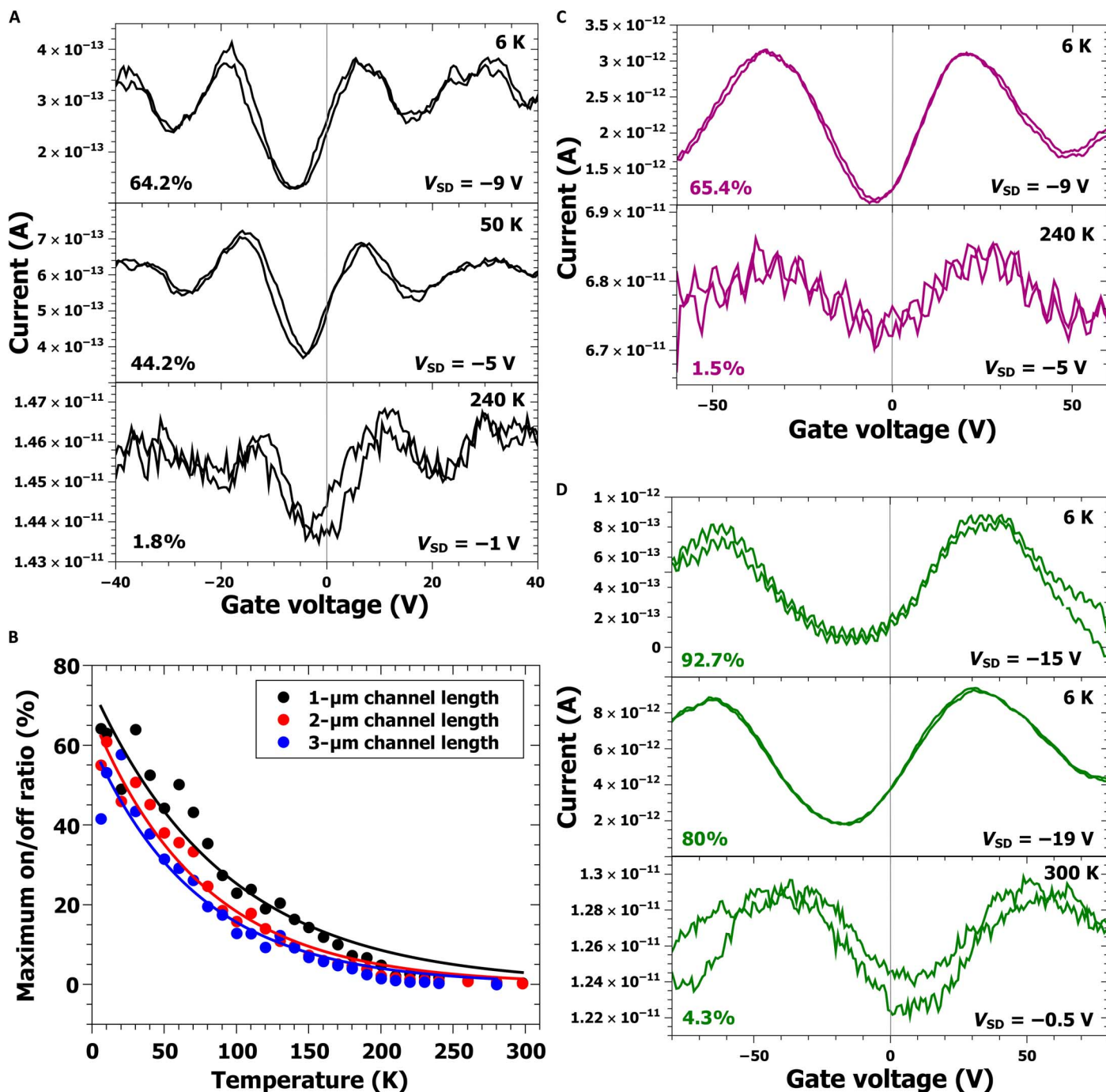
or gating geometry. Overall, the transport investigations support the monodispersity and supercrystallinity of the particle monolayer.

### Coulomb oscillations and tunability

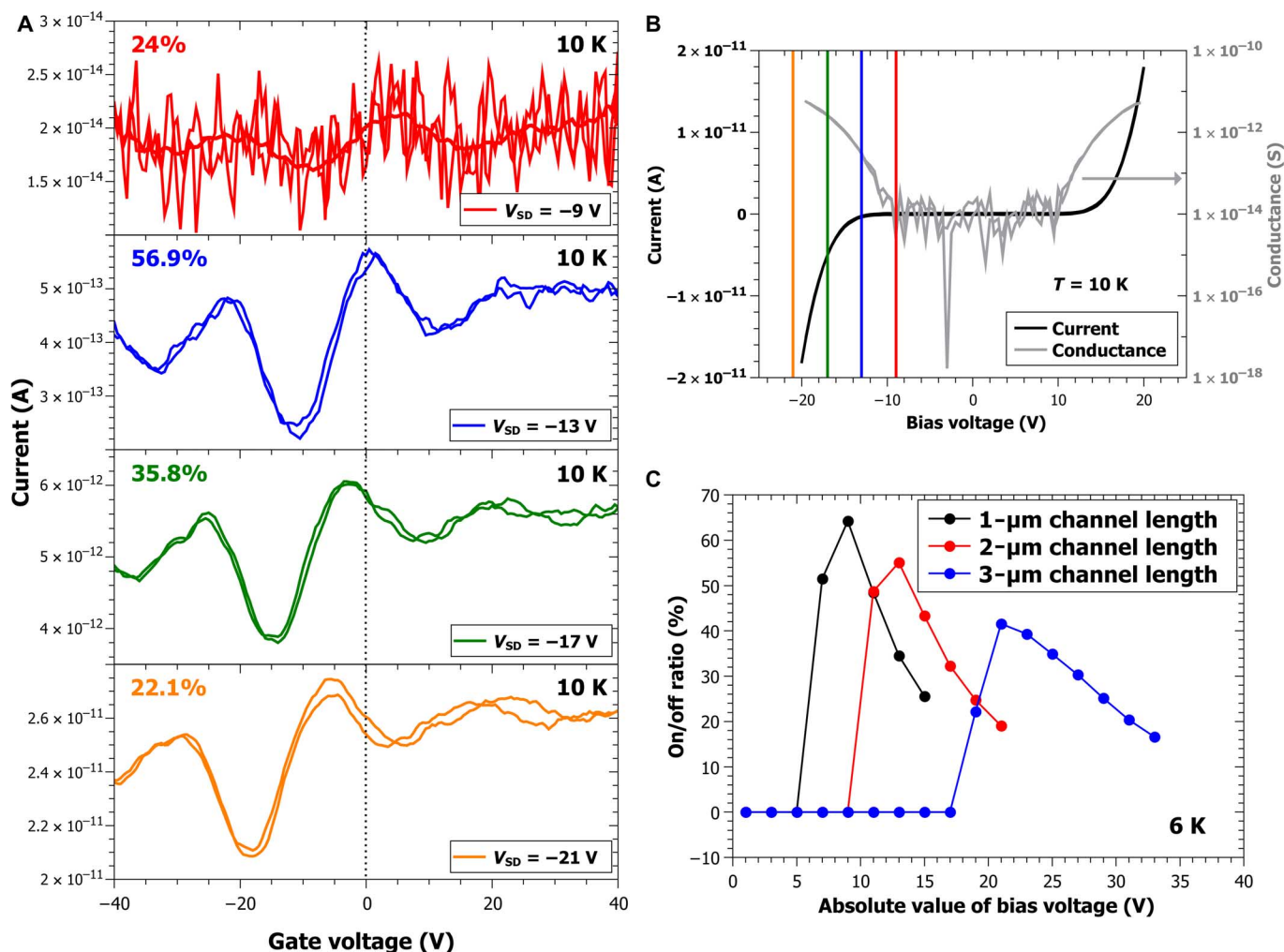
An additional gate electrode separated from the channel by a dielectric layer allows for influencing the transport capacitively. Tuning the gate voltage results in a field effect that continuously shifts the energy levels

of the particles. Whenever neighboring energy states align, tunneling is most probable, and a maximum current flow can be observed. Exemplary transfer characteristics are displayed in Fig. 3 (temperature dependency) and Fig. 4 (source-drain voltage dependency; a 2D pseudocolor

plot can additionally be found in fig. S6). They show these periodic Coulomb oscillations with, for extended particle systems, unprecedented regularity and clarity (32, 33). To quantify the results, we calculated the relative on/off ratios between main minimum and neighboring maximum



**Fig. 3. Temperature dependency.** (A) The local back gate induces clear oscillations with reproducing features during a bidirectional gate-voltage sweep for different temperatures. (B) The highest found on/off ratios for three different channel lengths measured with a locally gated sample decay exponentially with increasing temperature. (C) A sample gated with the global substrate shows very periodic oscillations with similar percentages. (D) Smaller nanoparticles (2.3 nm diameter) lead to higher on/off ratios and more pronounced oscillations at higher temperatures. Minor changes in bias voltage adjust the current and render the curve perfectly smooth. Even above room temperature, clear Coulomb oscillations are visible and efficiencies of 5% could be reached. A larger distance to the gate electrode and smaller particles elongate the oscillation period from an average of 24.3 V (A) to 54.7 V (C) and 96.9 V (D). Note that the origin of the higher frequency oscillations seen in (C) and (D) is not entirely clear; they are assumed to result from some external influence and only appeared from time to time.



**Fig. 4. Voltage dependency.** (A) Oscillations induced by a local gate for different bias voltages at a temperature of 10 K. For the lowest shown voltage, an averaged curve has been added as a guide to the eye. The shape and distinctive features of each curve are reproducing at different bias voltages. A shift in minimum position can very well be seen. (B) Corresponding output characteristics and differential conductance plot. The colored lines mark the bias voltages displayed in (A). (C) The bias voltage dependency of the on/off ratio illustrates the different stages of electrical transport in the system. The maximum performance is reached around the threshold voltage.

(for consistency always on the right side) and these can be found in the graphs as percentages.

Sweeping the potential of a local 0.5- $\mu$ m-wide back gate below a 1- $\mu$ m-long conduction channel resulted in the transfer characteristics shown in Fig. 3A. It can be seen that the current clearly oscillates and that there is only little noise and hysteresis between the two sweep directions. All main features reproduce for different temperatures, and there is one pronounced minimum, which will be discussed in detail later. At low temperatures, on/off ratios of about 70% could be observed with the medium-sized particles. With increasing temperature, the electrons' energy and thus the overall current are higher as more and more conduction paths are accessible for transport. However, this multitude of conduction paths cannot be simultaneously switched as efficiently because of statistical variations of threshold voltage and Coulomb energy. This results in a quasi-constant background current, and the relative on/off ratio decreases exponentially with temperature as displayed in Fig. 3B. Three devices are compared, differing only in channel length. Although the differences are not large, there is a tend-

ency that shorter channels yield higher on/off ratios with the same-sized back gate.

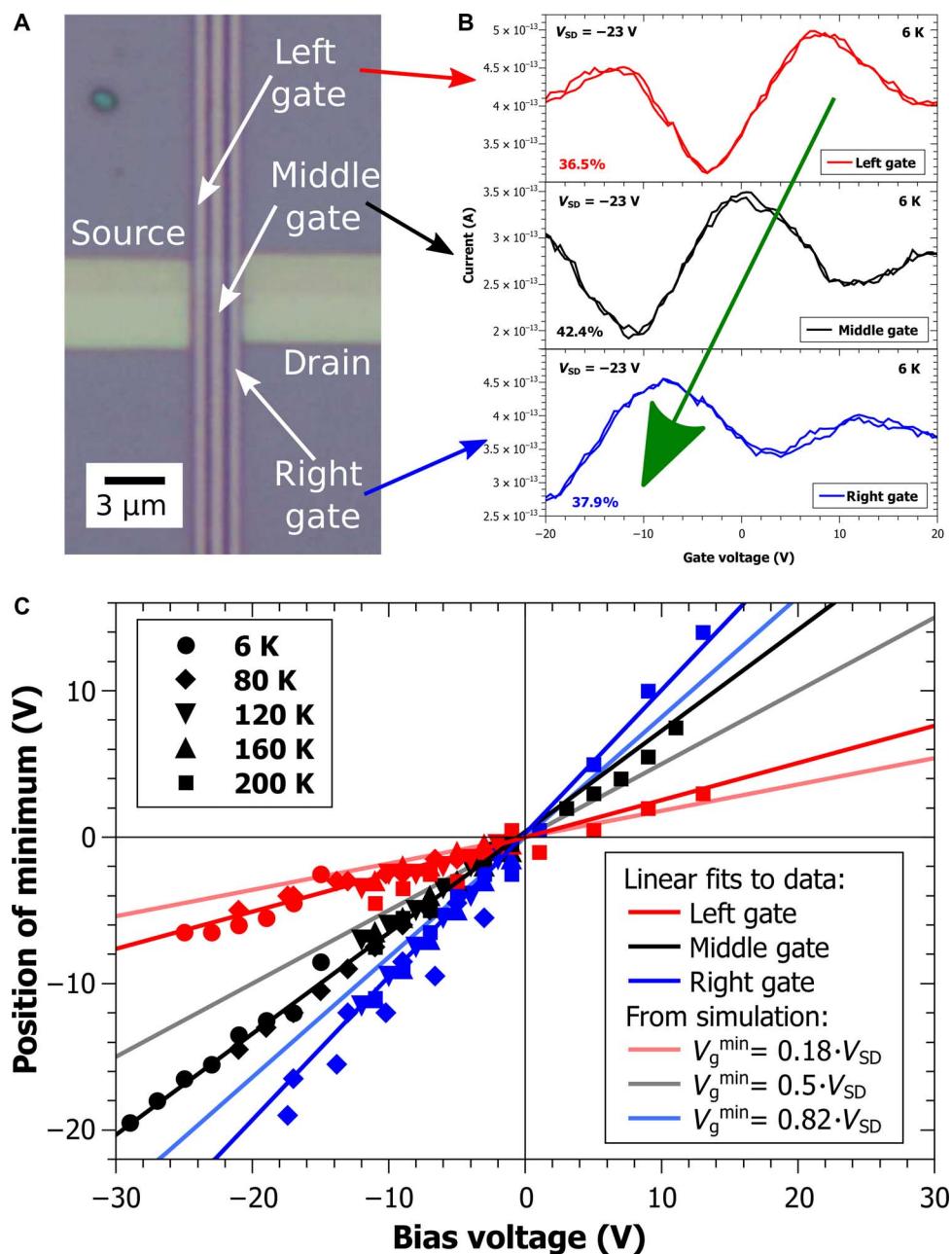
Using the global silicon substrate as back gate, which stretches over the whole channel length in contrast to the local gate, yields Coulomb oscillations with even higher regularity and periodicity (Fig. 3C). The transfer characteristics appear sinusoidal, and there is no pronounced main minimum anymore. The device performance is about the same compared to the local gate case (Fig. 3A). Note that clear oscillations can be observed at temperatures of up to at least 240 K. Beyond that, both gating geometries show evidence of room temperature Coulomb oscillations. However, the obtained percentages were negligible.

To improve the device performance even further, the adjustability of the system can be used in yet another way. By decreasing the particle size, the self-capacitance is lowered and the charging energy is increased. Hence, Coulomb oscillations become more pronounced and remain visible up to higher temperatures. Figure 3D shows results from a device with smaller nanoparticles of 2.3 nm diameter gated again by the global substrate. At the lowest measured temperature of 6 K, a relative on/off

ratio of more than 90% could be obtained. Moreover, sinusoidal oscillations were clearly detectable at room temperature. The on/off ratio was six times higher than observed before, and values of up to 5% could be measured.

Further investigations demonstrate the influence of the bias voltage on the oscillations at 10 K. This can exemplarily be seen in Fig. 4A; the chosen voltage values are also marked by the vertical lines in the respective output characteristics in Fig. 4B (more data are given in fig. S7). The lowest displayed bias voltage of  $V_{SD} = -9$  V is clearly still inside the blockade regime. Nevertheless, weak oscillations can be seen in the red

curve. Although the signal is very noisy, this shows how the gate's field effect can lower the energy barrier so that current flow is periodically possible. Increasing the bias voltage enables more conduction paths. However, similar to the effect of a high temperature, the paths all vary slightly in their energy-level configuration and cannot be blocked simultaneously, leading to lower relative on/off ratios. Thus, the best ratios were obtained for bias voltages near the threshold voltage, as depicted in Fig. 4C. This subfigure summarizes exemplary device performances at a temperature of 6 K for different bias voltages. For each curve, there is a blockade regime, then an onset of oscillatory behavior,



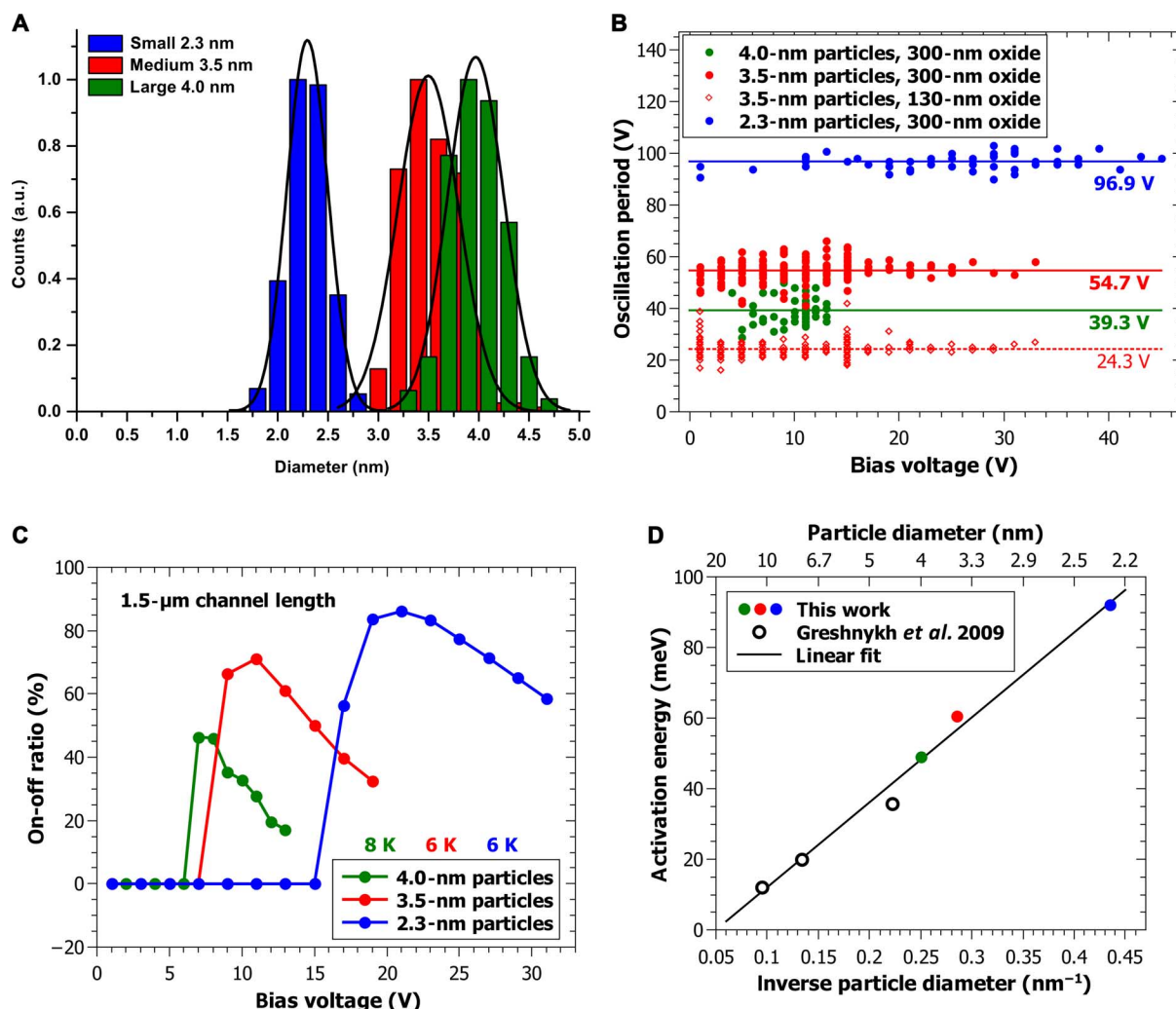
**Fig. 5. Oscillation shift.** (A) Optical micrograph of a device with a 3- $\mu$ m-long channel and three parallel 500-nm-wide gate electrodes. (B) Each gate can be used to induce oscillations. At the same bias voltage, the main features are shifted. (C) Plotting the position of the minimum with respect to the applied bias voltage yields a linear correlation that is independent of temperature. Each gate position corresponds to a certain slope. The measured data roughly match the simulated slopes but reveal a shift in the lithographic alignment.

a maximum of the on/off ratio around the threshold voltage, followed by a slow decrease. The longer the conducting channel, the higher the respective oscillation onset and threshold voltage because the total bias is distributed over more particles. Again, it can be seen that the best performance was observed in the device with the shortest channel.

The four oscillations shown in Fig. 4A are shifting toward more negative gate voltages as the applied bias changes in the same direction. In the pronounced main minimum of the gate-voltage sweep, not only the energy-level configuration is most unfavorable for transport, but also the overall electric-field gradient between the three contacts is disadvantageous for elastic tunneling. Depending on the exact location of the gate electrode under the channel and on the gate width, a simple model (32) suggests that this main minimum is at  $V_{\text{gate}} \approx V_{\text{SD}}/2$ . At this gate voltage, the linear potential gradient is least disturbed and remains very unfavorable for resonant tunneling so that the current is minimal. The potential gradient between source and drain is simulated and evaluated for two different gate positions based on this model in fig. S8. To

investigate this effect in more detail, we fabricated structures with three parallel narrow local back gates, as can be seen in the optical micrograph in Fig. 5A. They have the same width, are separated from the channel by the same amount of dielectric and therefore only differ in their lateral placement, and influence the transport along the channel from different positions.

Applying the same bias voltage to source and drain and using each one of the gates alone while leaving the other two uncontacted (floating) results in a shift in the oscillations of about 16 V depending on the used gate's position (Fig. 5B). It is now possible to use the gate voltage corresponding to the oscillation minimum as a marker for the shift and plot it for different bias voltages (Fig. 5C). Independent of the temperature, there is a linear correlation between the gate voltage at the minimum and the applied bias voltage. It was found that the slope of this line can be directly connected to the gate's position. The theoretical slopes from simple simulations (fig. S8) are also given in Fig. 5C. Although they are all somewhat smaller than those obtained from the data,



**Fig. 6. Particle size comparison.** (A) Size distributions including Gaussian fits. a.u., arbitrary units. (B) Period of oscillation for different channel lengths, temperatures, and bias voltages. The horizontal lines mark the average value for each set of data points. The period seems to be only dependent on particle size and separation between channel and gate electrode. With these two parameters, it can be tuned over a wide voltage range. (C) Bias voltage dependent on/off ratios at low temperature for the three sizes. The best performance can be obtained by using small particles. (D) The activation energy is inversely proportional to the particle diameter. The results of this work supplement the data from another publication (37).

the difference is about the same for all three lines. This leads to the conclusion that there might have been a shift in the lithography process, and the gates are positioned slightly more to the right than intended. Note that these gates are still quite wide compared to the channel. The correlation between gate placement and oscillation is valuable for applications: The position of a narrow local gate can be directly used to shift the resulting transfer characteristics or, vice versa, the shift can be used to determine the gate position. For an application that requires the working voltage to be at a certain point of the oscillation, for example in the minimum for current stabilization, this offers an easy, accessible tuning mechanism to adapt the functionality even if the bias voltage is given. Extending the study with the three gates to different channel lengths (fig. S9) confirms the tendency that a better effect is obtained if the gate stretches over a larger fraction of the channel and adds that a gate electrode has no influence anymore if it is placed right next to the channel underneath the gold leads. In summary, the above observations show that depending on the focus of the desired application, one could either use a narrow local gate to tune the position of the oscillations or a wide gate that covers the whole channel area to maximize the on/off ratio.

The device geometry determines many of the system's characteristic properties; however, the nanoparticle diameter has an even stronger influence. The usage of wet-chemically synthesized colloidal nanoparticles makes small sizes and precise tailoring possible. In Fig. 6A, the size distributions of three batches of particles are shown. They range between 2.3 nm and 4.0 nm: The smaller the particles, the higher the self-capacitance and thus the charging energy. This can be seen directly in the transfer characteristics: Sweeping the gate voltage from one maximum to the next means adding the required energy to overcome the corresponding Coulomb energy gap. The period of oscillation becomes longer for smaller particles. This correlation can be seen very well in Fig. 6B. For each particle size, a point cloud of oscillation period measurements is given. It was found that the period did not depend on channel length, temperature, or bias voltage, which is in line with the theory. The horizontal lines mark the average value also given as a number on the right. There is also a second influence on the periodicity: The thickness of the dielectric material, meaning the separation of channel and gate electrode, determines the strength of the electric field in the nanoparticles. A thinner dielectric strongly increases the field effect and also shortens the oscillation period (compare red lines in Fig. 6B). Although a thin dielectric also improves device performance, it still needs a certain thickness to act as barrier against leakage currents.

Increasing the Coulomb energy by decreasing particle size also leads to more pronounced oscillations. They remain visible up to a higher temperature and exhibit higher on/off ratios. Figure 6C summarizes device performance at the lowest measured temperature for the three particle sizes. Simply changing from the medium particles to the small ones increased the on/off ratio by about 15% and enabled sinusoidal room temperature oscillations. As discussed in the beginning of this work, the system's activation energy can be determined from the Arrhenius plots of the differential zero-bias conductance. Being strongly related to the Coulomb energy itself, the activation energy is also proportional to the inverse particle diameter. In Fig. 6D, the three data points from particles investigated in this work extend the data from previous work (31). The linear fit matches the relation very well and confirms the expected behavior.

Fabricating the gate electrode before particle deposition leaves the particle array that constitutes the conducting channel undisturbed and the Coulomb oscillations can be observed with little interference. Therefore, the influence of different gating techniques and parameters could be reliably investigated. It has been observed that even the sputter-

deposited silicon dioxide underneath the particles can influence the transfer characteristics. Charging of trap states in the gate dielectric leads to some reversible hysteresis effects if the applied gate voltage is very high (see fig. S3). For later applications, some form of sealing will be required. It was found that polymer systems such as resist or wax do not interfere with the device's function (see fig. S4).

## CONCLUSION

We demonstrate a new device concept that allows for easy and industry-compatible fabrication and a high tunability to match the needs of transistors or future applications. The device parameters can especially influence working temperature and bias, as well as oscillation period and strength. The use of colloidal nanoparticles ensures the tunability of size and tunnel barrier width, monodispersity, very small capacitances and thus high Coulomb energies, and inexpensiveness. In the metal nanoparticle-based devices, Coulomb oscillations are observed with an unprecedented clarity and reliability, even at room temperature.

## MATERIALS AND METHODS

### Nanoparticle synthesis

CoPt nanoparticles were synthesized according to the work done by Shevchenko *et al.* (40) with some modifications. The mainly investigated particles were monodisperse with a diameter of  $(3.5 \pm 0.3)$  nm and a composition of 93.2% Pt and 6.8% Co [analysis of the (111) x-ray diffraction (XRD) peak by means of Vegard's law]. In the first step, 33 mg (0.084 mmol) of the platinum source platinum(II) acetylacetonate [Pt(acac)<sub>2</sub>] was dissolved in 4 ml of the solvent 1-octadecene. To assist in the reduction of the platinum, 131.9 mg (0.51 mmol) of 1,2-hexadecanediol was added. For shape and size control during particle nucleation and growth, 2.22 ml (6.72 mmol) of oleylamine and 0.44 ml (1.4 mmol) of oleic acid were added. The solution was heated to 80°C and stirred at this temperature for at least an hour under vacuum conditions. In the second step, 43.7 mg (0.126 mmol) of the cobalt source dicobalt octacarbonyl [Co<sub>2</sub>(CO)<sub>8</sub>] dissolved in 0.6 ml of 1,2-dichlorobenzene in an ultrasound bath was injected rapidly into the solution that has been heated to 160°C under nitrogen atmosphere. This so-called hot injection was followed by thermal decomposition of the cobalt precursor. The mixture immediately turned black and was continuously stirred at the injection temperature for 2 hours. During cool down, 5 ml of toluene was added. Then, the particles were washed three times through precipitation with 2-propanol and methanol, phase-separated in a centrifuge at 7000 revolutions per minute (4492g) for 4 min, and subsequently resuspended in toluene. In the final step, they were filtered through a polytetrafluoroethylene syringe filter with a porousness of 0.2 μm. Other particle sizes can be realized by altering the ratio of amine and platinum source. Various interparticle distances can be achieved by substituting longer- or shorter-chained ligands.

### Sample preparation

Doped silicon wafers of roughly 1 cm<sup>2</sup> with 500 nm of thermal oxide were used as substrates. For electrode preparation, they were covered with the positive resist poly(methyl methacrylate) (PMMA) (dissolved in chlorobenzol) via spin-coating. Standard electron-beam lithography [Quanta Scanning Electron Microscope (FEI) and CAD software ELPHY Plus (Raith)] and metal evaporation of 2 nm of titanium as an adhesion layer and 23 nm of gold created the electrodes and contact pads as well as markers to align the different layers during lithography.



As a gate dielectric, an isolating layer of about 130 nm of silicon dioxide was deposited via ion-beam sputtering [Precision Etching and Coating System model 682 (Gatan)] on top of the back-gate electrode, while the contact pad was protected with PMMA that could later be removed. In the last step, another resist mask was created, which defines the conduction channel as a stripe between source and drain electrode. Some samples did not have a local back gate but were measured using the global silicon substrate as a gate. In these cases, the pristine thermal oxide was only 300 nm thick.

### Langmuir-Blodgett method

Monolayer preparation was done using the Langmuir-Blodgett trough from KSV NIMA, as seen in fig. S2. The silicon substrates with readily fabricated electrodes were attached to a holder together with a transmission electron microscope (TEM) grid to investigate the film quality afterward. They were submerged into the subphase diethylene glycol. The particles were washed again, dissolved in toluene, and carefully brought upon the subphase surface with a microliter syringe (starting with 300 to 400  $\mu\text{l}$  of nanoparticles in toluene, as described under synthesis for a trough area of 120  $\text{cm}^2$ ). Because of the hydrophobic character of the ligand shell, they will stay on the surface and spread as far as possible while the solvent evaporates. With two computer controlled barriers, they were slowly forced together with a speed of 1 mm/min; the compression was monitored by in situ surface-pressure measurement. The slope changes of the resulting isothermal curve and empirical known values of the required difference in surface pressure enabled the estimation of the point of a densely packed monolayer without holes or overlap. The barriers kept the film at this pressure for 2 hours of relaxation time. Then, the substrates were carefully lifted and slowly broke the surface at an angle of 105°, which allowed collection of the particle film (30).

### Particle and sample characterization

The CoPt nanoparticles were investigated with a TEM after synthesis and film preparation. During each Langmuir-Blodgett process, a TEM grid was loaded with particles in parallel to the actual silicon substrates. From these micrographs, the particle diameter, size distribution, spacing, and monolayer quality were assessed. Exemplary batches of particles were measured with XRD to verify the crystal structure and composition. All throughout the fabrication process, the substrates, electrodes, resist masks, and the final sample were frequently checked with a light microscope. Furthermore, in exemplary samples, the heights of the electrodes and of the sputtered dielectric were measured with an atomic force microscope.

### Electrical characterization

The main measurements were carried out in a closed cycle cryogenic probe station (model CRX-4K, Lake Shore Cryotronics) with a Keithley 4200-SCS parameter analyzer. Characterization was done at residual gas pressures of  $10^{-6}$  mbar at room temperature to  $10^{-7}$  mbar at low temperature. The sample stage can be cooled down to 4.5 K. Contacting was done with up to six micromanipulatable needle probes with a 20- $\mu\text{m}$ -radius beryllium-copper tip. Another pin was connected to the sample stage and was used to apply a voltage to the substrate and thus to the global back gate. Signal transmission was achieved through tri-axial cables to the Keithley's source measure units. Currents down to about 100 fA can be detected, whereas everything below was considered as noise. For output characteristics, the gate was left floating or grounded, whereas the bias voltage was varied stepwise between  $\pm 20$  V. Simi-

larly, for transfer characteristics, the gate voltage was varied stepwise in different intervals depending on the sample with a fixed bias voltage. Parameter sweeps were always done bidirectionally to verify the results.

### SUPPLEMENTARY MATERIALS

Supplementary material for this article is available at <http://advances.sciencemag.org/cgi/content/full/3/7/e1603191/DC1>

text S1. Capacitance.  
fig. S1. Particle characterization.  
fig. S2. Monolayer preparation.  
fig. S3. Hysteresis.  
fig. S4. Capping.  
fig. S5. Transport mechanism.  
fig. S6. 2D plot of the device characteristics.  
fig. S7. Voltage-dependent oscillations.  
fig. S8. Position of minimum.  
fig. S9. Influence of gate position.  
References (41, 42)

### REFERENCES AND NOTES

1. L. Esaki, New phenomenon in narrow germanium *p-n* junctions. *Phys. Rev.* **109**, 603 (1958).
2. L. Esaki, Long journey into tunneling. *Proc. IEEE* **62**, 825–831 (1974).
3. M. H. Devoret, D. Esteve, C. Urbina, Single-electron transfer in metallic nanostructures. *Nature* **360**, 547–553 (1992).
4. K. Likharev, Single-electron devices and their applications. *Proc. IEEE* **87**, 606–632 (1999).
5. N.-H. Kaneko, S. Nakamura, Y. Okazaki, A review of the quantum current standard. *Meas. Sci. Technol.* **27**, 032001 (2016).
6. C. J. Gorter, A possible explanation of the increase of the electrical resistance of thin metal films at low temperatures and small field strengths. *Physica* **17**, 777–780 (1951).
7. R. I. Shekhter, Zero anomalies in the resistance of a tunnel junction containing metallic inclusions in the oxide layer. *Sov. Phys. JETP* **36**, 747 (1973) [translation from *Zh. Eksp. Teor. Fiz.* **63**, 1410 (1972)].
8. I. O. Kulik, R. I. Shekhter, Kinetic phenomena and charge discreteness effects in granulated media. *Sov. Phys. JETP* **41**, 308–316 (1975).
9. T. A. Fulton, G. J. Dolan, Observation of single-electron charging effects in small tunnel junctions. *Phys. Rev. Lett.* **59**, 109–112 (1987).
10. J. H. F. Scott-Thomas, S. B. Field, M. A. Kastner, H. I. Smith, D. A. Antoniadis, Conductance oscillations periodic in the density of a one-dimensional electron gas. *Phys. Rev. Lett.* **62**, 583–586 (1989).
11. H. van Houten, C. W. J. Beenakker, Comment on “Conductance oscillations periodic in the density of a one-dimensional electron gas.” *Phys. Rev. Lett.* **63**, 1893 (1989).
12. C. W. J. Beenakker, Theory of Coulomb-blockade oscillations in the conductance of a quantum dot. *Phys. Rev. B* **44**, 1646–1656 (1991).
13. R. Lavieville, F. Triozon, S. Barraud, A. Corna, X. Jehl, M. Sanquer, J. Li, A. Abisset, I. Duchemin, Y.-M. Niquet, Quantum dot made in metal oxide silicon-nanowire field effect transistor working at room temperature. *Nano Lett.* **15**, 2958–2964 (2015).
14. S. Lee, Y. Lee, E. B. Song, T. Hiramoto, Observation of single electron transport via multiple quantum states of a silicon quantum dot at room temperature. *Nano Lett.* **14**, 71–77 (2014).
15. S. J. Shin, C. S. Jung, B. J. Park, T. K. Yoon, J. J. Lee, S. J. Kim, J. B. Choi, Y. Takahashi, D. G. Hasko, Si-based ultrasmall multiswitching single-electron transistor operating at room-temperature. *Appl. Phys. Lett.* **97**, 103101 (2010).
16. K. Nishiguchi, A. Fujiwara, Y. Ono, H. Inokawa, Y. Takahashi, Room-temperature single-electron transfer and detection with silicon nanodevices. *IEDM Technical Digest* 199–202 (2004).
17. D. V. Talapin, J.-S. Lee, M. V. Kovalenko, E. V. Shevchenko, Prospects of colloidal nanocrystals for electronic and optoelectronic applications. *Chem. Rev.* **110**, 389–458 (2010).
18. C. de Mello Donegá, Synthesis and properties of colloidal heteronanostructures. *Chem. Soc. Rev.* **40**, 1512–1546 (2011).
19. D. L. Klein, P. L. McEuen, J. E. Bowen Katari, R. Roth, A. P. Alivisatos, An approach to electrical studies of single nanocrystals. *Appl. Phys. Lett.* **68**, 2574–2576 (1996).
20. M. G. Ancona, W. Kruppa, R. W. Rendell, A. W. Snow, D. Park, J. B. Boos, Coulomb blockade in single-layer Au nanocluster films. *Phys. Rev. B* **64**, 033408 (2001).
21. A. Bezryadin, C. Dekker, G. Schmid, Electrostatic trapping of single conducting nanoparticles between nanoelectrodes. *Appl. Phys. Lett.* **71**, 1273–1275 (1997).

22. J. Kane, M. Inan, R. F. Saraf, Self-assembled nanoparticle necklaces network showing single-electron switching at room temperature and biogating current by living microorganisms. *ACS Nano* **4**, 317–323 (2010).
23. K. I. Bolotin, F. Kuemmeth, A. N. Pasupathy, D. C. Ralph, Metal-nanoparticle single-electron transistors fabricated using electromigration. *Appl. Phys. Lett.* **84**, 3154–3156 (2004).
24. S. Kumagai, S. Yoshii, N. Matsukawa, K. Nishio, R. Tsukamoto, I. Yamashita, Self-aligned placement of biologically synthesized Coulomb islands within nanogap electrodes for single electron transistor. *Appl. Phys. Lett.* **94**, 083103 (2009).
25. K. Whitham, J. Yang, B. H. Savitzky, L. F. Kourkoutis, F. Wise, T. Hanrath, Charge transport and localization in atomically coherent quantum dot solids. *Nat. Mater.* **15**, 557–563 (2016).
26. R. Parthasarathy, X.-M. Lin, H. M. Jaeger, Electronic transport in metal nanocrystal arrays: The effect of structural disorder on scaling behavior. *Phys. Rev. Lett.* **87**, 186807 (2001).
27. C.-W. Jiang, I.-C. Ni, S.-D. Tzeng, C.-S. Wu, W. Kuo, Identification of Mott insulators and Anderson insulators in self-assembled gold nanoparticles thin films. *Nanoscale* **6**, 5887–5893 (2014).
28. I. Langmuir, The constitution and fundamental properties of solids and liquids. II. Liquids. *J. Am. Chem. Soc.* **39**, 1848–1906 (1917).
29. K. B. Blodgett, Films built by depositing successive monomolecular layers on a solid surface. *J. Am. Chem. Soc.* **57**, 1007–1022 (1935).
30. V. Aleksandrovic, D. Greshnykh, I. Randjelovic, A. Frömsdorf, A. Kornowski, S. V. Roth, C. Klinke, H. Weller, Preparation and electrical properties of cobalt-platinum nanoparticle monolayers deposited by the Langmuir-Blodgett technique. *ACS Nano* **2**, 1123–1130 (2008).
31. D. Greshnykh, A. Frömsdorf, H. Weller, C. Klinke, On the electric conductivity of highly ordered monolayers of monodisperse metal nanoparticles. *Nano Lett.* **9**, 473–478 (2009).
32. Y. Cai, J. Michels, J. Bachmann, C. Klinke, Metal nanoparticle field-effect transistor. *J. Appl. Phys.* **114**, 034311 (2013).
33. H. Lehmann, S. Willing, S. Möller, M. Volkmann, C. Klinke, Coulomb blockade based field-effect transistors exploiting stripe-shaped channel geometries of self-assembled metal nanoparticles. *Nanoscale* **8**, 14384–14392 (2016).
34. S. I. Lim, I. Ojea-Jiménez, M. Varon, E. Casals, J. Arbiol, V. Puntes, Synthesis of platinum cubes, polypods, cuboctahedrons, and raspberries assisted by cobalt nanocrystals. *Nano Lett.* **10**, 964–973 (2010).
35. Y. Cai, D. Wolfkühler, A. Myalitsin, J. Perlich, A. Meyer, C. Klinke, Tunable electrical transport through annealed monolayers of monodisperse cobalt–platinum nanoparticles. *ACS Nano* **5**, 67–72 (2011).
36. P. Y. Yu, M. Cardona, *Fundamentals of Semiconductors: Physics and Materials Properties* (Springer, ed. 4, 2010).
37. R. H. Fowler, L. Nordheim, Electron emission in intense electric fields. *Proc. R. Soc. Lond. A* **119**, 173–181 (1928).
38. N. F. Mott, Conduction in glasses containing transition metal ions. *J. Non-Cryst. Solids* **1**, 1–17 (1968).
39. A. L. Efros, B. I. Shklovskii, Coulomb gap and low temperature conductivity of disordered systems. *J. Phys. C* **8**, L49–L51 (1975).
40. E. V. Shevchenko, D. V. Talapin, A. L. Rogach, A. Kornowski, M. Haase, H. Weller, Colloidal synthesis and self-assembly of CoPt<sub>3</sub> nanocrystals. *J. Am. Chem. Soc.* **124**, 11480–11485 (2002).
41. M. Lundstrom, J. Guo, *Nanoscale Transistors: Device Physics, Modeling and Simulation* (Springer, 2006).
42. S. Chen, R. W. Murray, S. W. Feldberg, Quantized capacitance charging of monolayer-protected Au clusters. *J. Phys. Chem. B* **102**, 9898–9907 (1998).

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